

Monolithically integrated III-Sb CW super-luminal light emitting diodes on non-miscut Si (100) substrates

G. Balakrishnan, M. Mehta, M.N. Kutty, P. Patel, A.R. Albrecht, P. Rotella, S. Krishna, L.R. Dawson and D.L. Huffaker

Reported is super-luminescent emission under room-temperature, continuous-wave conditions from GaSb quantum-well-based light emitting diodes (LED), monolithically integrated on Si (100) substrates. The LEDs are realised with substrate growth temperature under 500°C for the entire process and the Si (001) substrate is non-miscut. The lattice mismatch at the AlSb/Si interface is accommodated by interfacial misfit dislocation arrays (IMF) resulting in low defect-density III-Sb material without thick metamorphic buffers. The devices are grown in etched trenches on the Si substrate to reduce anti-phase domains in the III-Sb. The *n*-Si substrate is contacted directly and thus current flows through the III-Sb/Si IMF interface. The diodes have extremely low leakage current density ($J_{leakage} < 0.2 \text{ A/cm}^2$) in the reverse bias (-10 V) and show very good diode characteristics but exhibit a slightly elevated forward resistance ($R \sim 27 \Omega$), likely to be because of the IMF. The super-luminal spectra peaks at $2.14 \mu\text{m}$ with maximum output power $\sim 0.125 \text{ mW}$.

Introduction: The integration of III-V materials with Si substrates has been a heavily researched topic in semiconductor sciences owing to the extensive technological benefits that would result from the merger of the two material systems [1–4]. Two decades of research have resulted in numerous devices such as lasers, detectors and transistors grown on Si substrates [5–8]. At present, however, the goal of III-V/Si integration is to achieve optoelectronic and electronic devices on CMOS platforms, with considerable effort placed towards obtaining reliable light sources on CMOS circuits [9]. Recent publications by several research groups have been focused on this very goal. While some researchers have demonstrated monolithic integration [10–12], Bowers and co-workers have achieved evanescent mode coupling using a wafer bonded integration approach that may be attractive for back-end processing since the necessary temperatures are very low ($< 400^\circ\text{C}$) [13]. A monolithic approach, however, may be more attractive for achieving more dense and more cost-effective integration solutions compared to wafer bonding if it circumvents thick metamorphic buffers and can be conducted on non-miscut (100) Si substrates.

In this Letter, we demonstrate monolithic III-Sb based super-luminal light-emitting diodes (LEDs) on Si(100) substrates using material that has extremely low domain density and threading dislocation density. The III-Sb material is nucleated on the Si substrate through an interfacial array of misfit (IMF) dislocations resulting in low threading dislocation densities without a thick buffer. We have already demonstrated optically pumped devices using such interfaces [14, 15]. However, the presence of anti-phase domains (APDs) and the resulting anti-phase boundaries (APBs) have prevented electrically injected devices. A well-demonstrated recipe to overcome these domains involves the growth of diodes on miscut (100) substrates, with the magnitude of the miscut ranging from 2 to 4° [2]. This approach is impractical, at least for short-term integration, since the miscut wafers render the process incompatible with CMOS. Blakely *et al.* have demonstrated that the steps in etch-confined non-miscut Si(100), when annealed under high temperatures, tend to migrate to the perimeter of the confinement resulting in step-free regions in the centre [16, 17]. In fact, complete step-removal on Si(100) can be achieved at temperatures $> 1200^\circ\text{C}$, which would then result in regions on the Si(100) surface where domain-free growth can be achieved. We have observed, however, that the growth of III-Vs in etch-confined Si regions alone, without any annealing, results in significant APD density reduction. The devices demonstrated in this Letter have been grown in trenches on Si(100) without the use of any pre-growth annealing procedures and with a trench width of $40 \mu\text{m}$ and a trench depth of $0.5 \mu\text{m}$.

Fabrication and results: Fig. 1 shows the structure of the device. The devices are grown on the stripe-etched Si(100) substrate that is *n*-doped at $1 \times 10^{18}/\text{cm}^3$. The device growth is preceded by an HF oxide removal etch to passivate the surface with H-atoms. The growth of the device is initiated with the 100 \AA thick AlSb nucleation layer

consisting of the interfacial misfit dislocations, grown at 500°C and doped at $5 \times 10^{17}/\text{cm}^3$. An *n*-doped ($5 \times 10^{17}/\text{cm}^3$) $\text{Al}_{0.45}\text{Ga}_{0.55}\text{Sb}$ cladding layer ($2.3 \mu\text{m}$), followed by an $\text{Al}_{0.35}\text{Ga}_{0.65}\text{Sb}$ waveguide layer ($0.3 \mu\text{m}$) completes the lower half of the device. The active region consists of six GaSb QWs (120 \AA) separated by 45 \AA wide $\text{Al}_{0.35}\text{Ga}_{0.65}\text{Sb}$ barriers. The active region is followed by an upper $\text{Al}_{0.35}\text{Ga}_{0.65}\text{Sb}$ waveguide ($0.3 \mu\text{m}$) and a *p*-doped $\text{Al}_{0.45}\text{Ga}_{0.55}\text{Sb}$ cladding layer ($1.5 \mu\text{m}$), doped $5 \times 10^{17}/\text{cm}^3$, followed by a highly *p*-doped GaSb contact layer ($1 \times 10^{19}/\text{cm}^3$). Deposited Ti/Au and Al/In metal forms the top *p*-contact and the bottom *n*-contact, respectively. The sample is then cleaved into bars with a 2 mm cavity length.

| | |
|--|--------------|
| GaSb contact 500 \AA <i>p</i> -doped $1 \times 10^{19}/\text{cm}^2$ | } $\times 6$ |
| $\text{Al}_{0.45}\text{Ga}_{0.55}\text{Sb}$ cladding $1.5 \mu\text{m}$ <i>p</i> -doped $5 \times 10^{17}/\text{cm}^2$ | |
| $\text{Al}_{0.35}\text{Ga}_{0.65}\text{Sb}$ WG $0.3 \mu\text{m}$ | |
| $\text{Al}_{0.35}\text{Ga}_{0.65}\text{Sb}$ barrier 45 \AA | |
| GaSb QW 120 \AA | |
| $\text{Al}_{0.35}\text{Ga}_{0.65}\text{Sb}$ WG $0.3 \mu\text{m}$ | |
| $\text{Al}_{0.45}\text{Ga}_{0.55}\text{Sb}$ cladding $2.3 \mu\text{m}$ <i>n</i> -doped $5 \times 10^{17}/\text{cm}^2$ | |
| IMF based AlSb nucleation 100 \AA <i>n</i> -doped $5 \times 10^{17}/\text{cm}^2$ | |
| Si substrate <i>n</i> -doped $1 \times 10^{18}/\text{cm}^2$ | |

Fig. 1 Device structure of monolithically grown III-Sb edge emitter on Si(100)

The diodes are pumped using a CW current source. The LED emission is collected using a multimode fibre and coupled to a monochromator/InSb detector. The fibre has a uniform spectral response at the wavelengths of interest. Fig. 2 shows the emission from the device at various injection currents, including 350, 450 and 500 mA. At $\sim 350 \text{ mA}$, the emission is primarily spontaneous. The FWHM of the emission envelope is 352 nm . At higher injection currents, both narrowing of the spectral envelope to $\sim 30 \text{ nm}$ and nonlinear emission intensity suggest stimulated emission. The peak emission wavelength is at $2.14 \mu\text{m}$ and the maximum spectrally integrated output power from the device at $I = 500 \text{ mA}$ is 0.125 mW . Poor facet quality along with near-threshold operation contributes to the highly multimode characteristic of this device. Facet polishing is expected to improve the device performance. We note that the emission wavelength is considerably longer than expected for GaSb QWs. This is attributed to the mismatch in thermal expansion coefficient between GaSb and Si leading to a slight tensile strain ($< 0.7\%$) in the QWs along with possible heating effects from the electrical injection process.

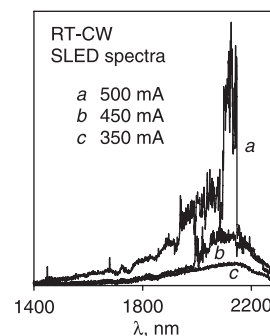


Fig. 2 Spectra from $50 \mu\text{m} \times 4 \text{ mm}$ device at injection currents of 350, 450 and 500 mA CW

Conclusions: We have demonstrated room temperature CW super-luminal LEDs monolithically grown on Si(100) substrates. The single domain III-Sb growth is achieved through growth of the devices in

etch-confined regions and low defect density is achieved through an IMF array-based growth mode. The reduced domain and defect density results in excellent diode characteristics. Electroluminescence spectrum at 2.14 μm is shown with maximum output power of 0.125 mW from the device.

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G. Balakrishnan, M. Mehta, M.N. Kutty, P. Patel, A.R. Albrecht, P. Rotella, S. Krishna, L.R. Dawson and D.L. Huffaker (Center for High Technology Materials, University of New Mexico, Albuquerque, NM 87106, USA)

E-mail: huffaker@chtm.unm.edu

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